

Investigations on (y)Ni_{0.8}Zn_{0.2}Fe₂O₄ (NZF) + (1-y)Ba_{0.9}Sr_{0.10}La_{0.001}Ti_{0.96}O₃ Composites

Vishal Khatri¹ · Vinay Kumar² · Sanjeev Kumar³ · Renu Rani⁴

¹Department of Physics, Nalanda College of Engineering, Chandi, Nalanda-803108, Bihar, INDIA. ²Department of Physics, COBS&H, CCS Haryana Agricultural University, Hisar-125004, Haryana, INDIA. ³Department of Metallurgical Engineering, IIT (BHU), Varanasi-221005, U.P. INDIA.

⁴P.G. Department of Physics, Magadh University, Bodhgaya-824234, Bihar, INDIA.

ABSTRACT

In the present paper, the properties of composites with composition $(y)Ni_{0.8}Zn_{0.2}Fe_2O_4$ (NZF) + $(1-y)Ba_{0.9}Sr_{0.10}La_{0.001}Ti_{0.96}O_3(y=0.00 \text{ and } 0.15)$ is presented. Solid state route was used to prepare the samples. Presence of both phases was confirmed by XRD pattern. Dielectric properties were measured with variation of frequency and temperature. Ferromagnetic and ferroelectric properties were also studied for composite samples.

© 2020 JMSSE · Indian Thermal Spray Society · Science IN. All rights reserved

Introduction

Studies on ferroelectric and ferromagnetic materials are technologically important research subjects, which led to new discoveries and many useful device applications because single phase materials have their limiting use in practical devices because magnetoelectric (ME) effect is considerably weak even at low temperature [1-2]. The alternatives are magnetoelectric composites because they exhibit large value of ME coefficient. Ferrite and ferroelectric are constituent phases of these composites. They can be exploited as waveguides, sensors, phase shifter, modulators and memory devices etc[3-6]. Limited data is available in the literature "on investigation of ferroelectric dielectric and properties of ferroelectromagnetic composites". In the present work, the addition of Sr in barium titanate can shift the Curie temperature towards room temperature and hence increase the room temperature dielectric constant. La substitution can further decrease the T_c which results in increase the room temperature dielectric constant. The Saturation magnetization of nickel ferrite increases with zinc content hence its magnetoelectric coefficient increases. MnO₂ was added to minimize the hopping mechanism $Fe^{2\scriptscriptstyle +} \leftrightarrow Fe^{3\scriptscriptstyle +}$ which results in an increase the resistivity of the samples[7-9]. In nickel ferrite, conductivity is mainly due to hopping between $Fe^{2+} \leftrightarrow Fe^{3+}$ In this paper, we present the detailed report on the properties of the prepared composite. They can be exploited as waveguides, sensors, phase shifter, modulators and memory devices etc [10].

Experimental

The individual phases (ferrite and ferroelectric) were prepared by conventional solid state reaction method. AR grade NiO, ZnO, Fe_2O_3 were used as raw materials for

Corresponding Author: Renu Rani, Tel: 9467928810 Email: renudhy@gmail.com

Contents lists available at http://www.jmsse.in/&http://www.jmsse.org/

ARTICLE HISTORY

 Received
 29-11-2020

 Revised
 02-12-2020

 Accepted
 02-12-2020

 Published
 06-12-2020

KEYWORDS

Composites Ferroelectric Polarization Magnetization Ferromagnetic

synthesis of ferrite phase. The mixing process was carried out by ball-milling using zirconia balls and distilled water as milling media. The slurry was dried and calcined in alumina crucible at 1000°C for 4 hrs. To the calcined powder, a small amount (0.5% by weight) of MnO_2 was added, ball milled and recalcined at 1000°C for 4 hours. The recalcined powder was then ball milled again and dried. The ferroelectric phase was prepared using the same route by using AR grade BaCO₃, SrCO₃ ,La₂O₃ and TiO₂ as raw materials and calcined at 1100°C. The composite was prepared by mixing 15% by wt. (NZF) phase to the (BSLT) (85% by wt.) phase. The mixing process was carried out by ball-milling using zirconia balls and distilled water as milling media. After drying, small amount of diluted PVA was added as binder and the pellets having 2-3mm thickness and 15 mm diameter were pressed using the uniaxial hydraulic press. The pellets were sintered at 1325°C for 4hrs with constant heating rate of 5°C/min. The structural characterization of the samples was carried out by using X-ray diffractometer using Cu Kα radiation $(\lambda = 1.541 \text{ Å})$. for electrical measurement the samples were then coated with silver paste and heated in an oven at 400°C for 1hr to ensure good ohmic contacts. The dielectric measurements were carried out as a function of frequency at different temperatures by using 4263B LCR meter interfaced to PC and programmable temperature chamber. P-E hysteresis loops were recorded at 20Hz for all the samples.

Results and Discussion

XRD pattern for constituent phases and their composite are shown in figure1. It can be observed that intensity of ferrite phase peak is very small because of low concentration of ferrite phase in composite sample. Calculated values of various parameters for all samples are given in table 1. To understand the microstructure of the



Table 1: Structural & dielectric parameters

		Ferroelectric	2	Ferrite	d_{exp}	d_{th}	Porosity	ε _{rt}	tanδ _{rt}	ϵ_{max}	$tan\delta_{max}$
У	a(Å)	c(Å)	c/a	a (Å)	(g/	cc)	(%)	at (35°C	,100 kHz)	(at10)0kHz)
0.00	3.9873	4.0027	1.0038	-	5.28	5.99	11.85	4305	0.068	17548	0.162
0.15	3.9880	4.0057	1.0044	8.3461	5.18	5.99	13.53	2393	0.039	3108	0.059

samples, SEM micrographs were recorded. SEM micrographs for both of the samples are shown in figure 2. It can be observed that the grain size of composite sample is small as compared to pure ferroelectric phase.

964



Figure 1: XRD pattern for a) Ferrite phase b) Composite c) Ferroelectric phase



b) y=0.15 Figure 2: SEM micrographs a) ferroelectric phase b) composite phase

At three different frequencies, the graph between dielectric constant and temperature is shown in figure 3. From figure it is observed that there are maxima for dielectric constant at a particular temperature called Curie temperature. All dielectric parameters for all samples are given in table1. From Fig.3 it can be observed that the dielectric constant for composite samples is less as compared to ferroelectric phase and also broadening of the peak is observed in composite samples. That may be due to incorporation of ferrite phase in pure ferroelectric phase that decrease the ferroelectric properties of the composites samples [11-12].



Figure 3: variation of dielectric constant with temperature for a) ferroelectric b) composite

At room temperature, the polarization- field (P-E) hysteresis loops and M-H hysteresis loop for composite samples are shown in figure 4. Presence of both loops indicating that composites are spontaneously polarized as well as spontaneously magnetization. The various ferroelectric parameters and magnetic parameters like) for all samples are shown in table 2. It is observed that as the content of the ferrite phase is added, all ferroelectric parameters decreases.



0.00 11.54 3.2 - 25.02	Y	P _r (μC/cm ²)	E _c (kV)	Ms(emu/g) Unpoled	$P_s(\mu C/cm^2)$
0.15 2.02 2.05 (.52 11.25	0.00	11.54	3.2	-	25.02
0.15 2.93 3.95 6.53 11.35	0.15	2.93	3.95	6.53	11.35



Figure 4: Ferroelectric and Ferromagnetic hysteresis loops for all samples

Conclusions

Ferroelectric rich ME composites have been prepared successfully by conventional solid state method. The dielectric peak for composite samples get broadened and depressed with addition of ferrite content. The observation of both P-E and M-H hysteresis loops in composite samples confirm the coexistence of ferroelectric and magnetic ordering in composites. Increase in P_r and P_s with increase in ferroelectric content and increase in M_r and M_s with increase in ferrite content shows that composites obey rule of mixtures.

References

- 1. J. Van, Suchetelene, Product properties: a new application of composite materials, *Philips Res. Rept.* 1972, 27, 28-37.
- 2. A. M. J. G. Van Run, D. R. Terrel and J. H. Scholing, An in situ grown eutectic magnetoelectric composite material, *J. Mater. Sci*, 1974, 9, 1710-1714.
- 3. Suryanarayana SV, Magnetoelectric interaction phenomenon in materials, Bull. Mater. Sci., 1994, 17 1259-1270.

- K Zhao, Chen K, Daj YR, Wan TG, Zhu JS, Effect of martensitic transformation on magnetoelectric properties Ni₂MnGa/PbZr_{0.5}2Ti_{0.48}O₃Ni₂MnGa/PbZr_{0.52}Ti_{0.48}O₃ composite Appl. Phys.Lett. 2005, 87, 162901.
- Rani R, Kumar P, Singh S, Juneja JK, Raina KK, Kotnala RK, Prakash C, Dielectric and ferroelectric properties of BST and NZF magnetoelectric composites, Int.Ferroelectrics, 2010, 122, 38-44.
- Jaffe B, Cook W , Jaffe H, Piezoelectric Ceramics ,Academic Press, London, 1971.
- Kumar P, Singh S, Juneja JK, Prakash C, Raina KK, Structural and ferroelectric properties of lanthanum modified BPZT ceramics, Materials Chemistry and Physics, 2011, 125 660-663.
- Dobal PS, Dixit A, Katiyar RS, Garcia D, Guo R, Bhalla AS, Phase transition in BST ceramics, Ferroelectric Lett., 2002, 29,1-10.
- Lokare S.A.; Patil D.R.; Devan R.S.; Choughle S.S.; Electrical conduction, dielectric behavior and magnetoelectric effect in BT-NiCoMnFe Composite, Material Research Bulletin, 2008,43, 326-331.
- Manglaraja R.U.; Ananthakumar S.; Manohar P.; Gnanam F.D.; Magnetic, electric and dielectric behaviour of NZF prepared through flush combustion method.J. Magn. Magn. Mater.; 2002, 253, 56-64.
- S. Upadhyay, D. Kumar, O. Prakash, Bull. Mater. Sci, Effet of composition on dielectric and electrical properties of SrLaTiCo System, 1996,19, 513-525.
- 12. C.M. Kanamadi, L.B. Pujari, B.K. Chougule, Dielectricbehavior and magnetoelectric effect in NCF-BPTZ composites, J. Magn. Magn. Mater,2005, 295, 139-144.

